

## Study Influence of a Magnetic Field on Thin Film of (CdTe) Preparation via PLAL

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### ABSTRACT

This work used pulsed laser ablation to influence a magnetic field to create CdTe (NPs) by using distilled water. According to X-ray diffraction results, the CdTe (NPs) product in a magnetic field is more crystalline. UV-Vis tests showed that CdTe (NPs) had an optical band gap shift from (1.75) eV to (2) eV. Energy-dispersive X-ray mapping confirmed the synthesis of CdTe (NPs) with a weight percentage ratio of 96% [Cd]/[Te]; the photoluminescence (PL) spectrum confirmed the emission of a solid and broad peak at (651.139). The current characteristics of the n-CdTe/p-SiHJ exhibit rectifying behavior when a magnetic field is applied, with the rectification factor rising from (48.351) to (62.105). The manufactured photodetector's spectral operating region was visible, with improved responsivity at 660 nm, increasing from 0.72 A/W to 0.84 A/W.

**Keywords:** CdTe thin films, Laser ablation, Magnetic field

### 1. INTRODUCTION

them apart from bulk materials, have drawn the attention of numerous researchers in recent times [1-4]. Because of their unique optical characteristics, CdTe nanoparticles (NPs), one of the II-VI semiconductor nanomaterials, have attracted the attention of many researchers [5-7].

CdTe, a highly absorbent P-type material, is ideal for thin-film solar cells due to its physiochemical characteristics. It includes an optical band gap that matches the AM1.5 spectrum at ambient temperature, resulting in an optical absorption coefficient of over  $5 \times 10^5 \text{ cm}^{-1}$ . [8]. There are many methods to preparation nanoparticles Nps with different size and shapes such as pulsed laser deposition PLD, sel-gel and pulse laser ablation in liquid. the pulse laser ablation in liquid. Compared to traditional methods, this one offers numerous advantages, including the capacity to monitor the size and features of the generated nanoparticles and the assurance that they are free of contamination; it is also used to make different kinds of nanostructured materials [9]. The production process is also straightforward, inexpensive, and does not need pricey vacuum chambers. It is straightforward to manage the ejected nanostructures as a suspension or powder since they are entirely collected in solutions and form a colloidal solution. Additionally, this alternative physical nanofabrication technique offered new avenues for material processing based on the PLA of solids in different liquids to create various unique materials, including nano diamond [10,17]. Here in, we report on the preparation of CdTe (NPs) under the influence of a magnetic field by pulsed laser ablation (PLAL) methods. characteristics study and Fabrication of CdTe nanoparticles /Si photodetector as demonstrated.

### 2. EXPERIMENTAL PROCEDURE

Colloidal CdTe nanoparticles were produced employing a Q-switched Nd: YAG laser with a wavelength of 1064 nm and a pulse duration of 7 ns figure (1) a picture of the device used during work. High-purity CdTe powder pellets from Sigma Aldrich were pressed with a 1.2 cm radius hydraulic compressor to make colloidal CdTe nanoparticles. In first step: To prepare colloidal CdTe nanoparticles, the CdTe target was placed in the bottom of a glass vessel filled with distilled water and exposed to focused Nd: YAG laser pulses at a laser frequency of 100 pulses with a laser energy density of 5.3 J/cm<sup>2</sup>. To focus the laser light on the target, a converging lens with a 10 cm length of focus was used. And second steps of the same preparation conditions. An external magnetic field of 20 mT was produced using four tiny external neodymium magnets. A Tesla meter and Gauss were used to measure the magnetic field (NVIS 621). to investigate how the magnetic field affects the characteristics of colloidal CdTe NPs while they are being laser blasted. An X-ray diffractometer (Philips model X/Pert Pro), an ELICO, SL174, Spectrophotometer, and a UV-Vis double beam spectrophotometer (Shimadzu, 00,912,901-S) were used to measure the optical absorption of colloidal CdTe nanoparticles in order to investigate photoluminescences. Morphology, particle size, and chemical composition were estimated using A Field Emission Scanning Electron Microscope (FE-SEM) (MIRA3 model-TESCAN). The substrate was a 1 cm<sup>2</sup> patch of mirror-like n-type silicon with an electrical resistance of 3-10  $\Omega \text{ cm}$ . To remove the natural oxide, the silicon substrates were first cleaned and etched off the silicon surface using diluted HF. After that, the silicon substrate was once more cleaned for an average of fifteen minutes using an ultrasonic bath (Cerry PUL 125) and distilled water. Following cleaning, photodetectors were created by applying a room-temperature, thin layer of

CdTe (about 0.7 μm) on a silicon substrate using a spin coater (Holmarc HO-TH-05).. Ohmic contacts have been added to the CdTe film using the thermal resistive technique. The CdTe/n-Si heterojunction's current-voltage properties were measured at room temperature in the dark and under

illumination. The photodetector's primary features, such as its spectral responsivity, were assessed using a calibrated electrometer, halogen lamp, DC power supply, and Jobin Yvon monochromator



Figure 1. Image of the device used in this work.

### 3. RESULT AND DISCUSSION

Figure 2 depicts the optical absorption of colloidal CdTe nanoparticles that were produced in both a 20 mT and no M-field. Smaller colloidal nanoparticles and a higher concentration of nanoparticles were made in the colloidal CdTe nanoparticles generated in a magnetic field during the ablation process.

In addition, their colour changed from brown to dark brown. It was supported by the photograph shown in Figure's inset Figure (2). Due to the increased nanoparticle concentration, CdTe NPs prepared with a magnetic field absorb lighter than those prepared without a magnetic field

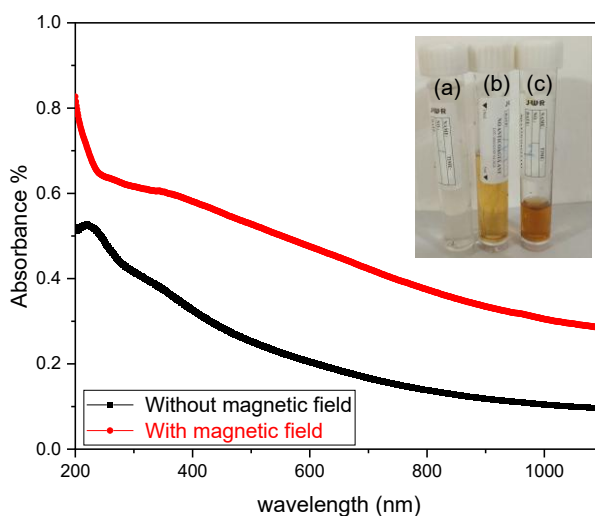


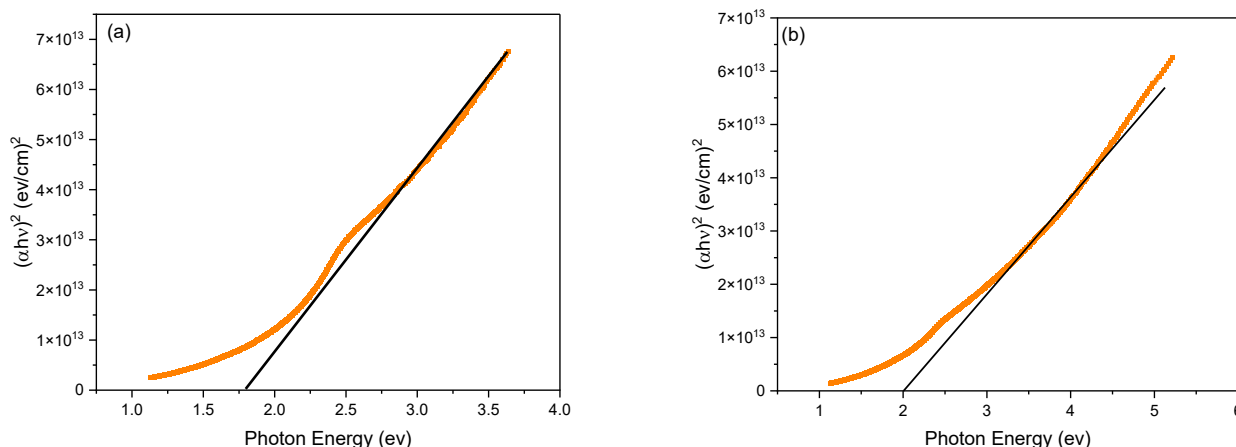
Figure 2. Optical absorption of CdTe colloid prepared without M-field(c) with 20 mT M-field. Inset is the photographs of colloids (a) water (b) without M-field(c) with 20 mT M-field.

As illustrated in Figure (3) for the CdTe sample prepared without M-field(c) with 20 mT M-field the energy band gap (Eg) of the allowed direct transmission was calculated using Tauc's equation [18,-21]:

$$\alpha h\nu = \beta (h\nu - E_g)^r \quad (1)$$

Where: α (absorption coefficient), hν (photon energy), β (band-tailing), r (It is a constant whose value depends on the type of electron transfer 1/2, 3/2, 2 and 3). The optical

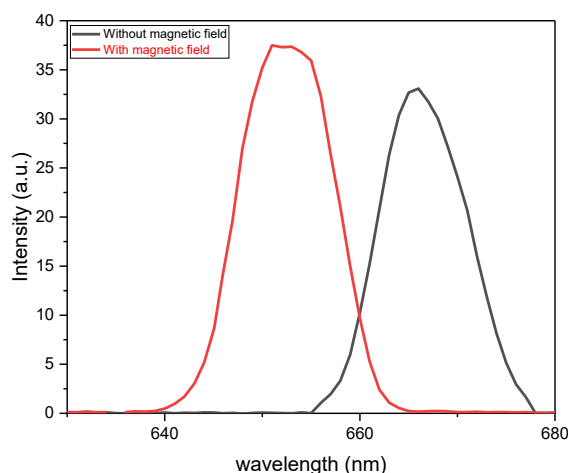
energy gap for CdTe NPs increased when a magnetic field was applied from (1.75) eV to (2.) eV. Due to the decrease in nanoparticle size [15,16] The duration and density of the plasma are increased by magnetic confinement caused by an external magnetic field. As a result, more energy is transferred from the plasma to the liquid and target, increasing the amount of material that is ablated. [22, 23]



**Figure 3.** Direct optical energy gap for CdTe NPs prepared (a) without M-field B=0(b) with M-field B=20mT.

The PL emission spectra of CdTe NPs are illustrated in Figure (4), prepared without M-field B=0 and M-field B=20mT. The PL emission of a CdTe sample prepared with M-field B=20mT exhibits a single distinct emission peak with high intensity and a wide FWHM compared to the PL emission of a CdTe sample prepared without a magnetic field. CdTe prepared with a magnetic field was centered at

(651.139) nm (1.904 eV), and the observed emission is displaced toward high energy or short wavelength. This results from the application of a magnetic field, which causes the size of the nanoparticles to decrease.



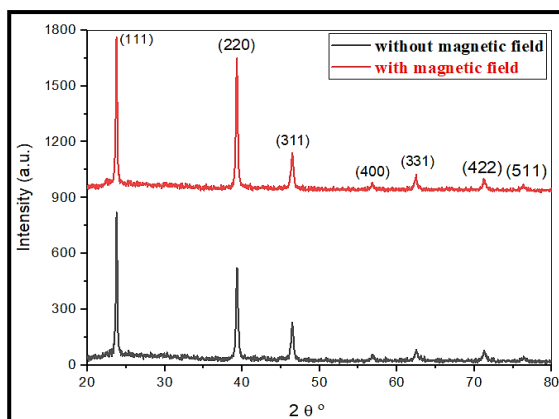
**Figure 4.** PL spectra of CdTe nanostructures colloidal synthesized without M-field B=0 and with M- field B=20mT.

The XRD pattern of CdTe nanoparticles generated at B = 0 and B = 20mT is displayed in Figure 5. Following the JCPDS card (no:15-0770).[17]. The XRD pattern of CdTe colloid Nanoparticles deposited on glass substrate showed diffraction peaks at  $2\theta = (23.7921^\circ, 39.3074^\circ, 46.4584^\circ, 56.7745^\circ, 62.4123^\circ, 71.2342^\circ, \text{ and } 76.3345^\circ)$  respectively, corresponding to the (111), (220), (311), (400), (331), (422), and (511) planes. The maximum intensity diffraction peak of the powder is also revealed by the XRD pattern at ( $2\theta = 23.7921^\circ$ ). To index the diffraction data, the cubic phase and associated lattice constant value ( $a = 6.466 \text{ \AA}$ ) were utilized [24]. The XRD pattern created using a magnetic field makes it evident that the CdTe peak's intensity increased. This outcome can be explained as improving the crystalline structure through increased crystallinity of the film material and reduced diffusion mobility of deposited

atoms, which enhances the growth and nucleation properties [25]. The average size of crystallites from preferred orientation (111) was determined For CdTe Scherrer's formula produces ( $D_{av}$ ) [26]:

$$D_{av} = (0.9 \lambda) / (\beta \cos\theta) \quad (2)$$

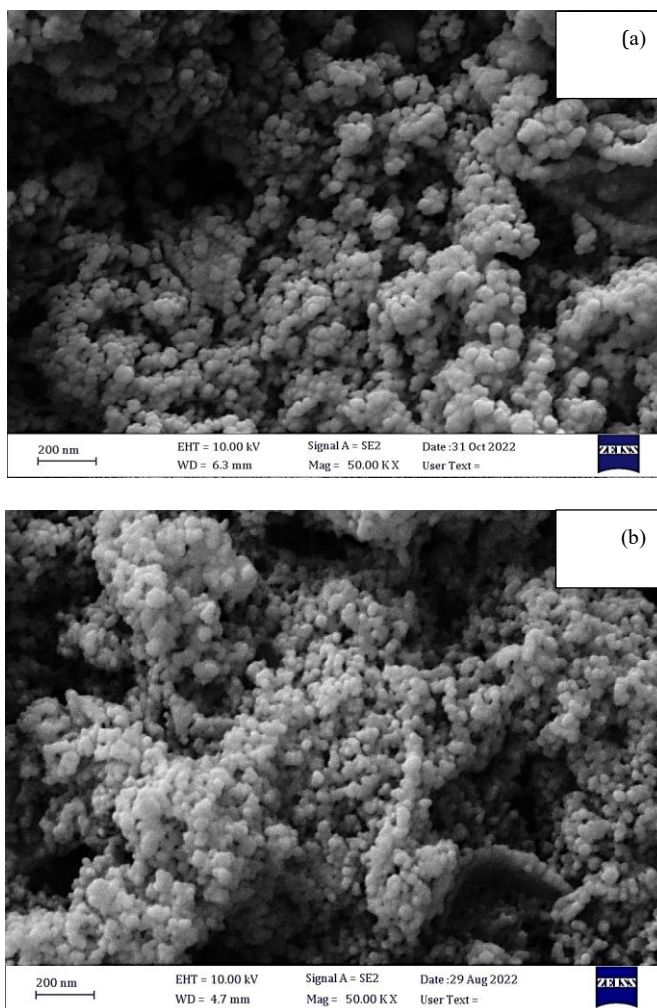
Where:  $\beta$  (full width at half maximum),  $\theta$  (diffraction angle), and  $\lambda$  (wavelength). The results of the calculations indicated that the application of a magnetic field caused the concentration of CdTe NPs to decrease, resulting in a decrease in crystallite size from (40.7231) nm to (36.605) nm when compared to the crystallite size without the magnetic field. [25,27].



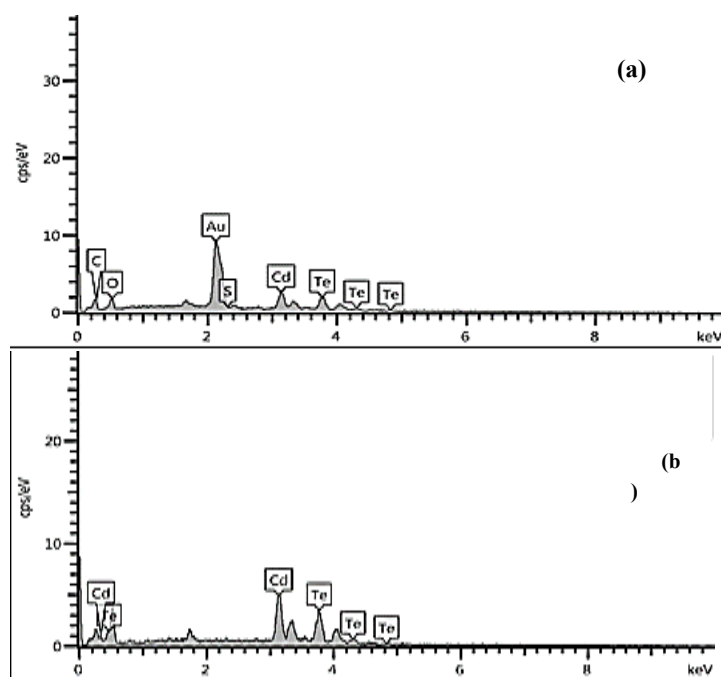
**Figure 5.** XRD patterns of CdTe nanostructures colloidal synthesized without M-field B=0 and with M- field B=20mT.

Figure (6b) exhibits CdTe nanoparticles generated in a magnetic field via laser ablation. as shown in the FESEM photos. When a magnetic field was absent, the film morphology in Figure (6a) was less homogeneous, dense, and compact. CdTe had an average particle size reduction from 26.17 nm to 20.92 nm. Figure 7 displays the EDS

spectra of a CdTe NPs film that was produced magnetically. The elements Cd and Te are seen in these spectra, along with glass substrate-related peaks O, C, and S. The table shows the weight % ratio [Cd]/[Te] for CdTe NPs produced in a magnetic field, as measured by EDS analysis (1).



**Figure 6.** FESEM images of CdTe nanostructures colloidal synthesized without M-field B=0 and with M- field B=20mT.



**Figure 7.** EDX spectra for CdTe NPs synthesised: (a) without M-field B=0 and (b) with M- field B=20mT.

**Table 1** Elements and weight percentage for CdTe nanostructures colloidal synthesized without M-field B=0 and with M- field B=20mT

CdTe	Wt % (Cd)	Wt % (Te)	Ratio Cd:Te
With magnetic field	49.2	50.8	0.96
Without magnetic field	40.8	52.5	0.77

CdTe NPs film synthesized with a magnetic field was determined to be p-type based on Hall Effect measurements. When a magnetic field is applied, CdTe NPs' electrical mobility improves and their electrical resistance decreases, as the table illustrates. (2). The increase in electrical mobility and decrease in resistivity when applying amagnetic field can be attributed to improved crystallinity, which promotes nanoparticle alignment and crystallization, as well as enhanced Charge Carrier Separation. The

magnetic field's alignment may allow for more efficient separation of electrons and holes, resulting in superior electrical characteristics. And Scattering at the grain boundary has been reduced. The magnetic field-induced alignment may reduce the number of grain boundaries or align them in a favorable direction, decreasing their adverse impact on electrical characteristics.

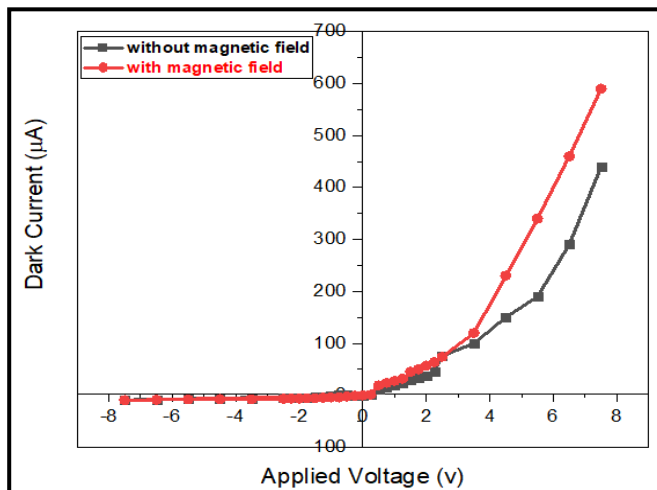
**Table 2** Electrical mobility and resistivity of CdTe NPs

CdTe	Electrical Mobility ( $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ ) * $10^3$	Electrical Resistivity ( $\Omega \text{ cm}$ )
Without magnetic field	8.97	3.98
With magnetic field	10.090	3.291

By applying a magnetic field, the dark I-V characteristics of the (p-CdTe/n-Si) heterojunction were studied, which extend the voltage range of -7.5 to +7.5 V, as shown in Figure 8. Because of the accelerated charge transfer and reduced resistance caused by the magnetic field, the heterojunction exhibits rectification features. The dark forward current for (p-CdTe/n-Si) increased from (440 $\mu\text{A}$ ) to (590 $\mu\text{A}$ ).

For p-CdTe/n-Si HJ, the rectification factor increases from (48.351) to (62.105) when applying a magnetic field. This improvement is attributable to improved junction properties (a reduction in structural defects and surface states).

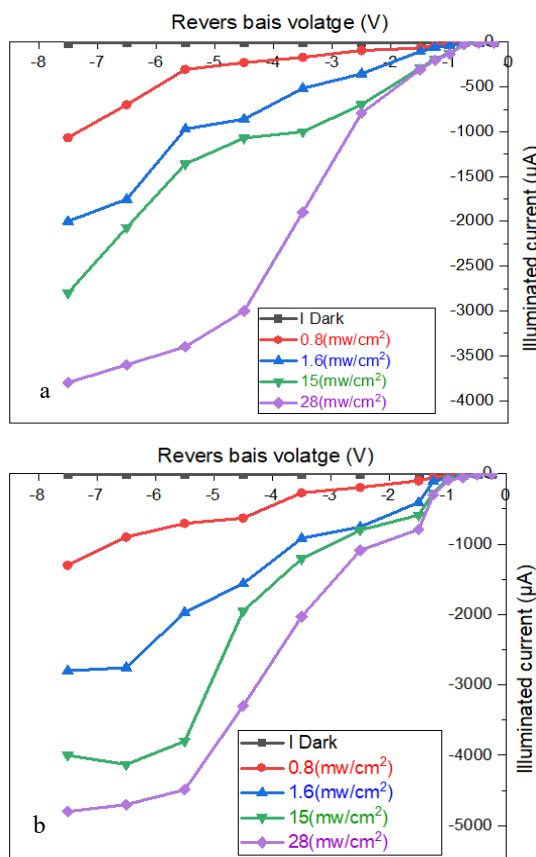




**Figure 8.** P-CdTe/n-Si heterojunction's dark I-V characteristic under forward and reverse bias.

Figure (9) depicts the I-V properties of the (p-CdTe/n-Si) heterojunctions in response to varying white light intensities and magnetic fields. When exposed to light, the photodetector's current increases due to the creation of an electron hole (e-h). As light intensity increases, photocurrent increases due to the growing number of photogenerated e-h couples. [28-34]. The photocurrent of the photodetector made in a magnetic field was higher than that of the photodetector made without one, as Figure (9b)

illustrates. The increased surface area and light absorption could be the cause of this rise in photocurrent once a magnetic field is applied. When manufactured with a magnetic field at 7.5 V bias, The photocurrent to dark current ratio ( $I_{ph}/I_d$ ) of the photodetector was 489.795 for (p-CdTe/n-Si). A decrease in particle size and an increase in the sensitive area are the reasons why applying the magnetic field increases the photodetector's photosensitivity. [33].



**Figure 8.** (I-V) Dark and illuminated characteristic of (p-CdTe/n-Si).

The spectral responsivity, which defines the ratio of output photocurrent to incident power, is a parameter used to measure the efficacy of any photodetector. Spectral responsivity measurement can be used to calculate the operating spectral region ( $\Delta\lambda$ ), peak response ( $\lambda P$ ), and responsivity value. The spectral responsivity ( $R_\lambda$ ) of p-CdTe/n-Si heterojunction was measured in the wavelength range of (360-910) nm with a bias of 0.65 V and calculated with the equation [35,36]:

$$R_\lambda = (I_{ph}/P_{in} \text{ or } V/P_{in}) \quad (3)$$

Figure 10 demonstrates how the application of a magnetic field affects the spectrum responsivity of a nanostructured (p-CdTe/n-Si) heterojunction photodetector. Two response peaks may be seen in photodetector responsivity data acquired both with and without a magnetic field. The first peak is located at 660 nm and is derived from the fundamental absorption of CdTe with responsivity (0.696 A/W). The second peak, on the other hand, is situated at 860 nm and is a part of the bulk silicon substrate's absorption edge. [16].

On the other hand, when a magnetic field is applied, the responsivity increases from 0.72 A/W to 0.84 A/W, with the first peak occurring at 660 nm. Additionally, the second peak, which has a responsivity of 1.2 A/W, can also be found at 860 nm.

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